

MoS₂-Based All-Purpose Fibrous Electrode and Self-Powering Energy Fiber for Efficient Energy Harvesting and Storage

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Here an all-purpose fibrous electrode based on MoS₂ is demonstrated, which can be employed for versatile energy harvesting and storage applications. In this coaxial electrode, ultrathin MoS₂ nanofilms are grown on TiO₂ nanoparticles coated carbon fiber. The high electrochemical activity of MoS₂ and good conductivity of carbon fiber synergistically lead to the remarkable performances of this novel composite electrode in fibrous dye-sensitized solar cells (showing a record-breaking conversion efficiency of 9.5%) and high-capacity fibrous supercapacitors. Furthermore, a self-powering energy fiber is fabricated by combining a fibrous dye-sensitized solar cell and a fibrous supercapacitor into a single device, showing very fast charging capability (charging in 7 s under AM1.5G solar illumination) and an overall photochemical-electricity energy conversion efficiency as high as 1.8%. In addition, this wire-shaped electrode can also be used for fibrous Li-ion batteries and electrocatalytic hydrogen evolution reactions. These applications indicate that the MoS₂-based all-purpose fibrous electrode has great potential for the construction of high-performance flexible and wearable energy devices.

planar or bulky structures, which cannot fully meet the urgent demands of portable, flexible, and wearable applications. Therefore, the fabrication of fibrous energy devices has attracted enormous attentions recently.^[10–21] However, so far the performances of fiber-shaped energy devices, such as DSSCs (Table S1, Supporting Information)^[22–29] and supercapacitors (Table S2, Supporting Information)^[30–41] are still lower than the corresponding planar devices, because it is difficult to find appropriate fibrous electrodes with high electrochemical activity and stability.

Fibrous carbon materials, including carbon nanotube fibers, graphene fibers, carbon fibers (CFs), as well as their hybrids, have been considered as a class of promising electrode materials for fibrous energy harvesting and storage devices, owing to their good conductivity, strong stiffness, high tensile strength, and light weight.^[42–48]

However, most of these carbon-based electrode materials are with intrinsically limited electrochemical activities, which significantly restrict their actual performances in fibrous energy devices. For instance, Zou and co-workers reported a fibrous DSSC constructed from bare CF-based counter electrode only showing a conversion efficiency of $\approx 1\%$.^[47] By decorating CoNi₂S₄ nanoribbons on the CF electrode, the photovoltaic conversion efficiency was enhanced to $\approx 7\%$, but still lower than that of the planar DSSC.^[47] In the aspect of supercapacitors, Lee and co-workers fabricated solid-state coaxial fibrous supercapacitors using all-carbon electrodes in polymer gel electrolyte, which retained only 46% of the initial capacitance as the scan rate was increased from 2 to 200 mV s⁻¹, indicating relatively low rate capability.^[48] To further improve the performances of fibrous energy devices, a promising route is to develop appropriate composite fibrous electrodes based on carbon and other complementary active materials with high electrochemical activity and efficiency.

On the other hand, it is of great importance to integrate energy conversion and storage functions into a single device, which can remove unnecessary charging circuits and external electrical connections, and reduce the size and cost. It is noteworthy that the integrated energy devices with low weight and compact size are particularly suitable for portable and wearable electronic applications. For examples, it has been reported that

1. Introduction

The energy and environmental crisis has become a vital problem threatening human society. To design energy harvesting and storage devices with high-efficiency and eco-friendliness is of great importance.^[1–3] In the past decades, lots of energy devices have been developed, such as dye-sensitized solar cells (DSSCs), supercapacitors, Li-ion batteries (LIBs), and fuel cells.^[4–9] However, most of existing energy devices are with

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planar solar cells can be stacked with supercapacitors or secondary batteries to simultaneously achieve photoelectric conversion and electrochemical energy storage (as shown in Table S3, Supporting Information).^[49–56] Nevertheless, the integrated energy devices with conventional planar structure cannot fulfill the demands of flexibility and elasticity for portable devices. Recently, several attempts have been made to develop wire-shaped integrated energy devices by connecting a fibrous DSSC and a fibrous supercapacitor, termed as “power fiber”, “energy wire” or “energy fiber” (see the bottom of Table S3, Supporting Information).^[39,57–59] However, up to now the performances of energy fibers are still relatively low, because it is difficult to design a multifunctional electrode with high efficiencies for both energy harvesting and storage. Therefore, it is desirable to design an all-purpose electrode with excellent performances in versatile energy devices, and the self-powering energy fibers.

In this study, we demonstrate an all-purpose fibrous electrode based on MoS₂ that can be simultaneously employed in various energy harvesting and storage applications. This wire-shaped coaxial electrode was prepared by the growth of ultrathin MoS₂ nanofilms onto TiO₂ nanoparticle coated carbon fiber (CF@TiO₂@MoS₂). The high electrochemical activity of MoS₂ and good conductivity of CF synergistically lead to the remarkable performances in fibrous DSSCs and fibrous supercapacitors. Furthermore, a self-powering energy fiber with integrated functionalities of DSSC and supercapacitor is developed by using the CF@TiO₂@MoS₂ electrode as an essential component. As a result, this energy fiber exhibited very fast photocharging rate and a high overall photochemical-electricity energy conversion efficiency. Besides, the MoS₂ based electrode also shows good capability in fibrous LIBs and electrocatalytic hydrogen evolution reactions (HER).

2. Results and Discussion

The structure of coaxial fibrous CF@TiO₂@MoS₂ electrode is shown in Figure 1. For the preparation, a strand of commercially

available CFs with high conductivity and flexibility (Figure S1, Supporting Information) was treated in TiCl₄ aqueous solution, and followed by hydrothermal treatment in a mixed solution containing Na₂MoO₄ and C₂H₅NS (as detailed in the Experimental Section of Supporting Information). In this way, a layer of TiO₂ nanoparticles were grown on CFs as a bifunctional film to provide nucleation sites for growing MoS₂ nanofilms and to enhance the performances of CFs by improving the surface area and catalytic activity. Following, ultrathin MoS₂ nanofilms were grown on TiO₂ nanoparticles coated CFs. The intention behind employing MoS₂ nanofilms as the active material can be ascribed to the following three reasons: (1) Due to the 2D structure, the MoS₂ nanofilms show ultrathin thickness and large surface area;^[60,61] (2) Because there are many active sulfur sites on the surface and edges, the MoS₂ nanofilms reveal good catalytic activities;^[60,61] (3) The MoS₂ nanofilms exhibit high energy storage capacity.^[62,63] Based on these favorable features, the fibrous CF@TiO₂@MoS₂ electrode was found to be very efficient for various energy conversion and storage applications, including DSSCs, supercapacitors, LIBs, and HER, as illustrated in Figure 1. In addition, a self-powering energy fiber integrated by a fibrous DSSC and a fibrous supercapacitor was also fabricated by using a single CF@TiO₂@MoS₂ electrode, exhibiting high efficiencies of both solar-to-electricity conversion and electrochemical energy storage.

The structural features of pristine CF and CF@TiO₂ electrode were characterized by scanning electronic microscopy (SEM), as shown in Figure S2 of the Supporting Information. It should be noted that the decoration of TiO₂ nanoparticles on CF is very important for the further coating of MoS₂ nanofilms because it provides necessary nucleation sites for the growth of MoS₂. Without the treatment of TiCl₄ solution, very few MoS₂ nanofilms can be grown on CF, as shown in Figure S3 of the Supporting Information. We also found that the growth of MoS₂ nanofilms on CF@TiO₂ is much better than that on slightly oxidized CFs (Figure S4, Supporting Information). Moreover, the optimization of processing time in the TiCl₄ solution was explored, as shown in Figures S2 and S5a of the

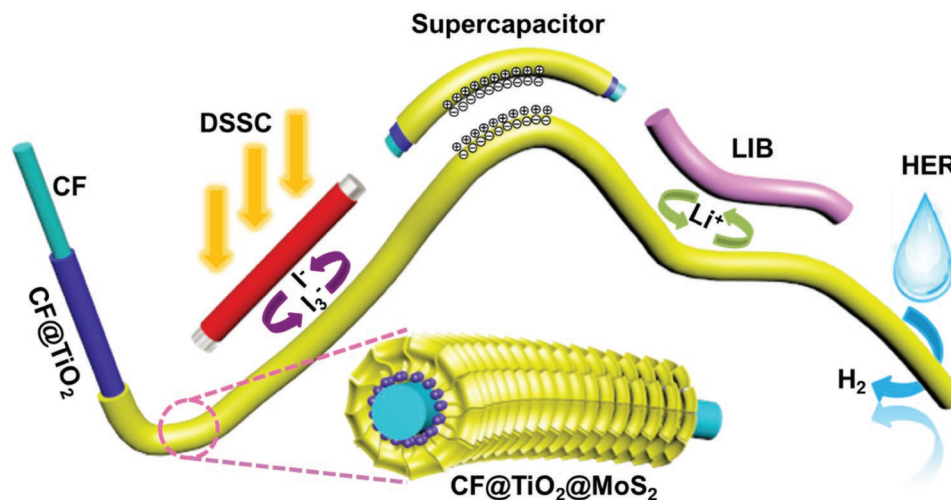


Figure 1. Schematic illustration of the structure of coaxial fibrous CF@TiO₂@MoS₂ electrode and its versatile applications for energy harvesting and storage, including fiber-shaped DSSCs, supercapacitors, LIBs, and flexible HER electrode.

Supporting Information. Similarly, the optimal synthesis condition of MoS₂ nanofilms in the mixed solution containing Na₂MoO₄ and C₂H₅NS was investigated as well, as shown in Figure 2a–c and Figure S5b of the Supporting Information. The SEM images of fibrous CF@TiO₂@MoS₂ electrodes obtained from the optimized synthesis condition at low and high magnifications (Figure 2a–c) revealed that the surface is uniformly covered by densely packed and wrinkled MoS₂ nanofilms. Figure 2d shows a SEM image and corresponding energy dispersive X-ray spectrometry (EDX) elemental mappings of the cross-section of a CF@TiO₂@MoS₂ electrode. The core of coaxial electrode is occupied by C element, while Ti, O, Mo, and S elements are evenly distributed in the outer shell, indicating that the CF is covered by the MoS₂ nanofilms with the help of TiO₂. Transmission electron microscopy (TEM) was also performed on as-prepared CF@TiO₂@MoS₂ electrode, which further revealed that the CF was fully covered by MoS₂ nanofilms (Figure 2e–g). Figure 2f shows most of the MoS₂ nanofilms are with very thin thickness. Clear lattice fringes can be observed in few-layered MoS₂ nanofilms (Figure 2g). The distance between two adjacent planes is ≈0.63 nm, which is in accordance with the interplanar spacing of (002) plane of hexagonal MoS₂ crystalline structure.^[64,65]

X-ray photoelectron spectroscopy (XPS) analysis of CF@TiO₂@MoS₂ electrode was carried out (Figure S6, Supporting Information). The survey XPS spectrum (Figure S6a, Supporting Information) is consistent with the above result of EDX. Figure S6b–f of the Supporting Information present the high-resolution XPS spectra of C 1s, Ti 2p, O 1s, Mo 3d, and S 2p regions; the values of binding energies are listed in Table S4 of the Supporting Information. These data are in accordance with the characteristics of TiO₂ and MoS₂. The Raman spectra

of CF@TiO₂@MoS₂ electrode are given in Figure S7 of the Supporting Information, which confirms the existence of carbon, TiO₂, and MoS₂.^[66–68] Thermogravimetric analysis (TGA) was carried out on CF@TiO₂ and CF@TiO₂@MoS₂ electrodes, respectively, as shown in Figure S8 of the Supporting Information. The residue of CF@TiO₂ after TGA measurement is TiO₂ (≈0.7%). Owing to the oxidative reaction of MoS₂ to MoO₃ under air atmosphere, the residues of CF@TiO₂@MoS₂ after TGA measurement are TiO₂ and MoO₃. The weight percentage of the residues of CF@TiO₂@MoS₂ after TGA measurement is ≈6.9%. Therefore, the content of MoS₂ in CF@TiO₂@MoS₂ was calculated to be ≈6.89 wt%.

Fibrous DSSCs were constructed by using CF@TiO₂@MoS₂ to replace traditional platinum (Pt)-based counter electrode. Figure 3a shows a schematic diagram of a fibrous DSSC based on CF@TiO₂@MoS₂ counter electrode. The working electrode is a Ti wire covered with dye-sensitized TiO₂ nanotube array (Figure 3b). Figure S9a of the Supporting Information shows the TiO₂ nanotube array is vertically attached on the surface of the Ti wire and possesses a uniform alignment. Raman spectrum of the Ti microwire covered with TiO₂ nanotube array was also measured, as shown in Figure S9b of the Supporting Information, which is in accordance with the previous literature and consistent with anatase phase.^[67] Figure 3c displays that the as-prepared fibrous DSSCs are very flexible to be bent into a circle or a heart-shape knot, and also can be wound around a glass rod. In principle, the overall power output of fibrous DSSCs is only limited by its length, which can be easily raised by increasing the length of the wire-shaped electrodes. Figure 3d shows a fibrous DSSC based on CF@TiO₂@MoS₂ with a length of 17 cm.

Cyclic voltammetry (CV) was employed to investigate the catalytic activity of as-produced electrodes toward tri-iodide

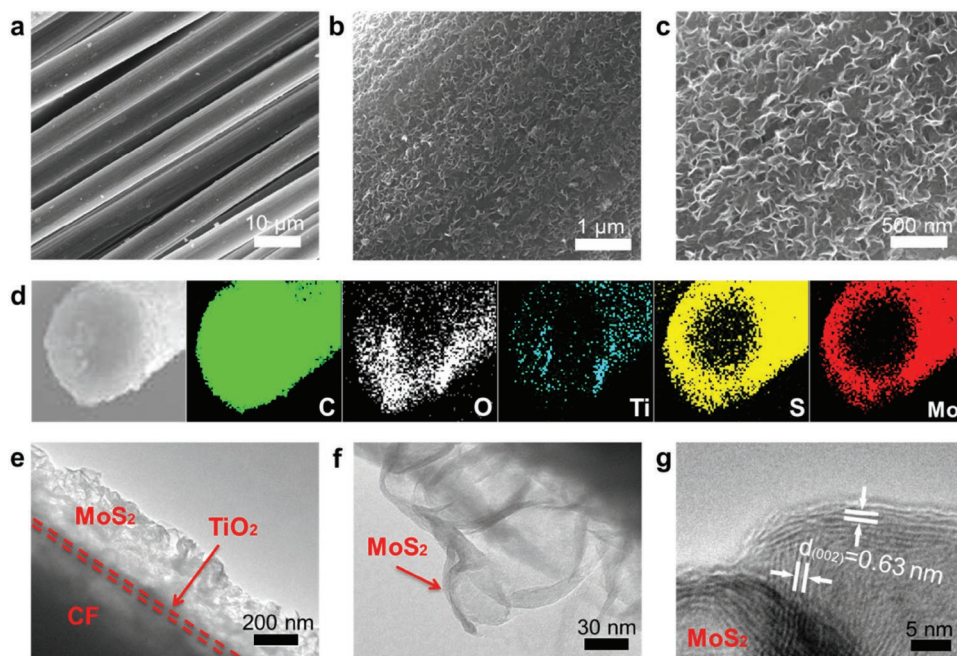


Figure 2. Structural characterizations of fibrous CF@TiO₂@MoS₂ electrode. a–c) SEM images of CF@TiO₂@MoS₂ electrode at different magnifications. d) SEM image and corresponding EDX elemental mapping of the cross-section of a CF@TiO₂@MoS₂ electrode. e–g) TEM images of CF@TiO₂@MoS₂ electrode. The lattice distance of 0.63 nm in (g) is assigned to the (002) planes of MoS₂.

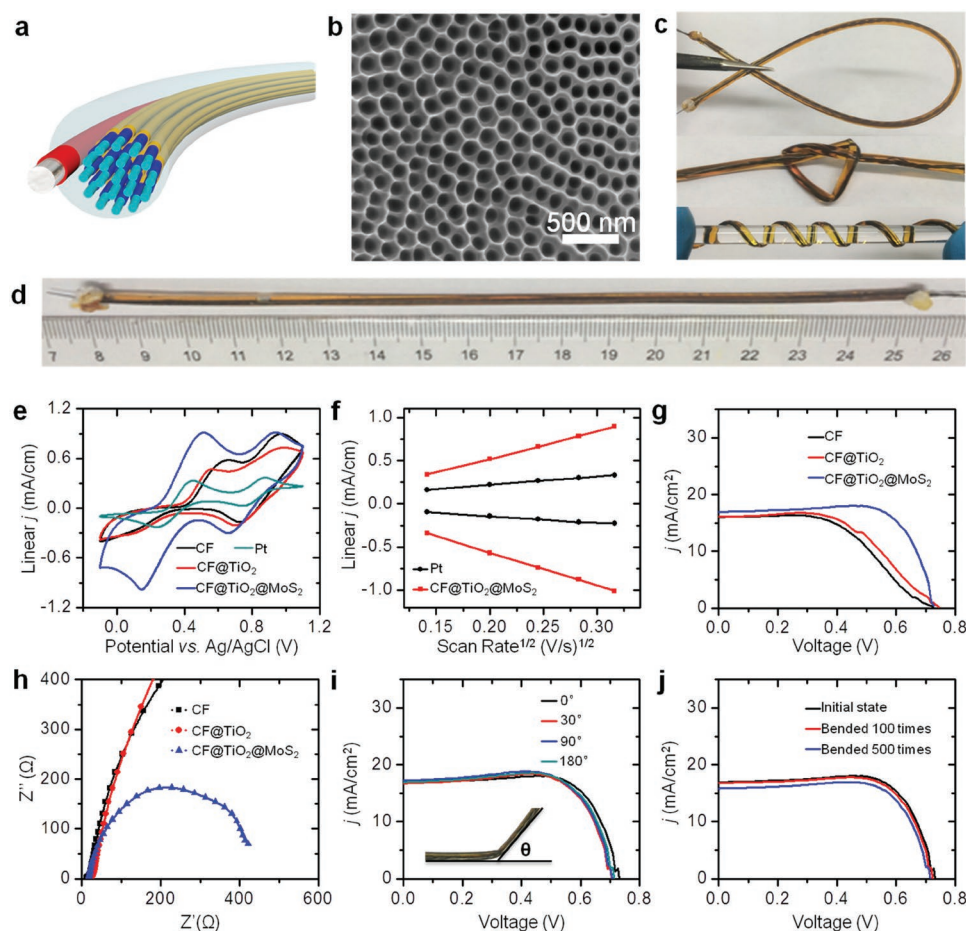


Figure 3. Fibrous DSSCs based on CF@TiO₂@MoS₂ electrode. a) Schematic illustration of a fibrous DSSC with CF@TiO₂@MoS₂ as counter electrode and dye-sensitized TiO₂ nanotube array coated Ti wire as working electrode. b) SEM image of TiO₂ nanotube array coated on the surface of Ti wire based working electrode. c) Optical images of fibrous DSSCs bended into various shapes. d) Optical image of a 17 cm long fibrous DSSC in the straight form. e) CV curves of pristine CF, CF@TiO₂, CF@TiO₂@MoS₂, and Pt counter electrodes at a scanning rate of 100 mV s⁻¹ toward I⁻/I₃⁻ redox couple, respectively. f) The peak current density of the left peak pairs as a function of the square root of the scan rates for Pt and CF@TiO₂@MoS₂ electrode. g) *J*-*V* plots and h) Nyquist plots of fibrous DSSCs based on pristine CF, CF@TiO₂, and CF@TiO₂@MoS₂ counter electrodes, respectively. i) *J*-*V* plots of fibrous DSSCs based on CF@TiO₂@MoS₂ counter electrode under different bending angles ($\theta = 0^\circ, 30^\circ, 90^\circ$, and 180° , respectively). j) *J*-*V* plots of fibrous DSSCs based on CF@TiO₂@MoS₂ counter electrode at initial state and after different bending-unbending cycles at $\theta = 30^\circ$.

reduction. Figure 3e shows the CV curves of the pristine CF, CF@TiO₂, CF@TiO₂@MoS₂, and Pt microwire counter electrodes toward I⁻/I₃⁻ redox couple. Two pairs of redox peaks were shown in these CV curves. The left peak pair represents the redox transition between I₃⁻ and I⁻, while the right peak pair represents I₂ and I₃⁻ transition. The electrocatalytic activity of the counter electrode material in the fibrous DSSC is directly related to the left oxidation–reduction peaks. Typically, the cathodic and anodic peaks on the left can be assigned to the reduction of I₃⁻ and the oxidation of I⁻, respectively.^[69] Clearly, the CF@TiO₂@MoS₂ electrode exhibits the highest electrocatalytic activity among the four samples. Moreover, the cycling stability of CF@TiO₂@MoS₂ based counter electrodes was measured by CV curves at a scanning rate of 100 mV s⁻¹, as shown in Figure S10 of the Supporting Information. The CV curve of CF@TiO₂@MoS₂ based counter electrode after 100 cycles almost completely overlaps with its initial CV curve. It indicates that the CF@TiO₂@MoS₂ based counter electrode has good stability during long-term tests. Figure S11 of the Supporting

Information shows the CV curves of CF@TiO₂@MoS₂ and Pt microwire at various scan rates. Based on the data in Figure S11 of the Supporting Information, the anodic and cathodic peak current densities of the left peak pairs as a function of the square root of the scan rates were plotted (Figure 3f). The peak current density of fibrous CF@TiO₂@MoS₂ electrode is much larger than that of the Pt microwire, suggesting that CF@TiO₂@MoS₂ electrode is a promising candidate for replacing Pt microwire counter electrode in fibrous DSSCs.

The performances of as-prepared fibrous DSSCs were systematically studied. Figure 3g shows the *J*-*V* plots of fibrous DSSCs based on pristine CF, CF@TiO₂, and CF@TiO₂@MoS₂ counter electrodes, respectively. The corresponding photovoltaic parameters are listed in Table S1 of the Supporting Information. The CF@TiO₂@MoS₂ based fibrous DSSCs display an open-circuit voltage (*V*_{OC}) of 0.74 V, a short-circuit current density (*J*_{SC}) of 16.95 mA cm⁻², and a high fill factor (FF) of 0.75, resulting in a very high conversion efficiency (η) of 9.5%. In contrast, the fibrous DSSCs with pristine CF or CF@TiO₂

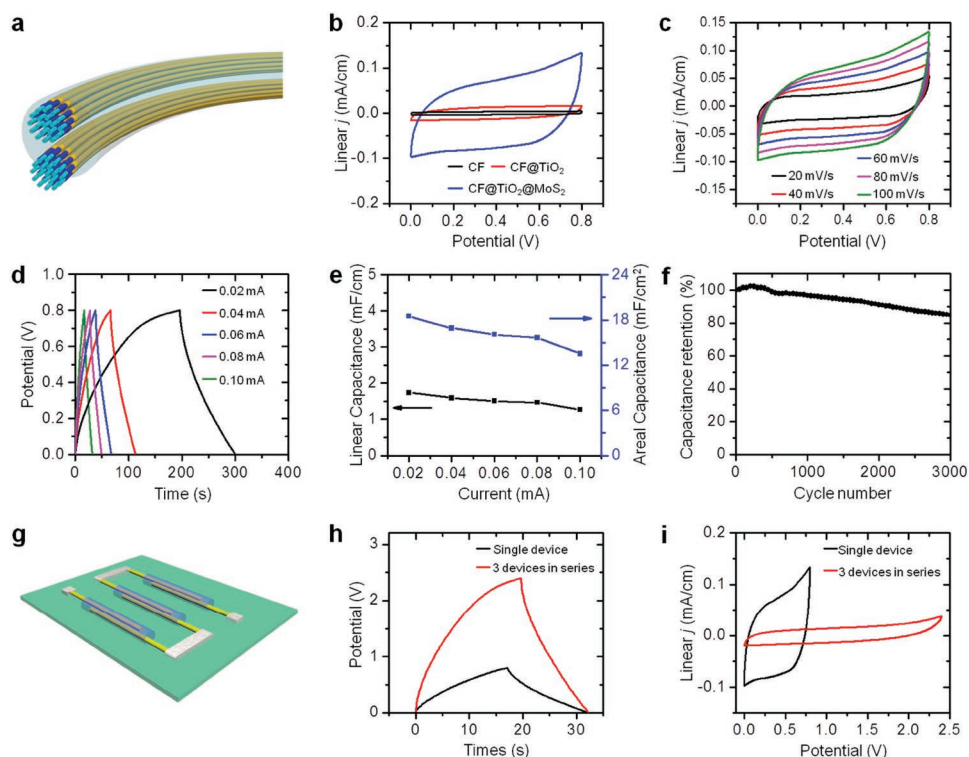


Figure 4. Fibrous supercapacitors based on CF@TiO₂@MoS₂ electrodes. a) Schematic illustration of a solid-state symmetric fibrous supercapacitor with CF@TiO₂@MoS₂ electrodes and PVA/H₃PO₄ gel electrolyte. b) CV curves of fibrous supercapacitors based on pristine CF, CF@TiO₂, and CF@TiO₂@MoS₂ electrodes at a scan rate of 100 mV s⁻¹, respectively. c) CV curves, d) galvanostatic charge/discharge curves, and e) linear and areal capacitances of CF@TiO₂@MoS₂ based fibrous supercapacitors, respectively. f) Performances of CF@TiO₂@MoS₂ based fibrous supercapacitors for 3000 cycles. g) Schematic diagram of an integrated high-voltage supercapacitor comprised of three in series-connected CF@TiO₂@MoS₂ based fibrous supercapacitors. h) Galvanostatic charge/discharge curves at a current of 0.10 mA and i) CV curves at a scan rate of 100 mV s⁻¹ of the integrated high-voltage supercapacitor and an individual fibrous supercapacitor based on CF@TiO₂@MoS₂ electrodes, respectively.

counter electrodes only display V_{OC} of 0.74 or 0.75 V, J_{SC} of 16.05 or 16.10 mA cm⁻², FF of 0.51 or 0.54, η of 6.0% or 6.5%, respectively. It should be noted that, to the best of our knowledge, the η of 9.5% achieved by the CF@TiO₂@MoS₂ based fibrous DSSCs is the highest if comparing with other fibrous DSSCs in previous reports (Table S1, Supporting Information).^[22–29] The dramatic increased FF and η of CF@TiO₂@MoS₂ based fibrous DSSCs is ascribed to the high electrocatalytic activity of MoS₂ and the rapid charge-transfer kinetic process occurring on the electrolyte-counter electrode interface. Figure 3h and Figure S12 of the Supporting Information show the Nyquist plots of the three kinds of fibrous DSSCs and the corresponding equivalent circuit, respectively, confirming the lowest charge transfer resistance (R_{ct}) of the CF@TiO₂@MoS₂ based fibrous DSSCs.

Moreover, we also tested the flexibility and stability of the fibrous DSSCs under bending deformation. Figure 3i shows the J - V plots of the CF@TiO₂@MoS₂ based fibrous DSSCs at different bending angles (θ). The corresponding photovoltaic parameters are listed in Table S5 of the Supporting Information. Figure 3i and Table S5 of the Supporting Information reveal that the η of the fibrous DSSC have no obvious change under different bending states. Furthermore, the J - V plots of CF@TiO₂@MoS₂ based fibrous DSSCs at initial state and after different bending-unbending cycles at $\theta = 30^\circ$ were measured

(Figure 3j), and the corresponding photovoltaic parameters are listed in Table S6 of the Supporting Information. After bending for 500 times, the η still can be maintained at 8.7% ($\approx 92\%$ retention of the initial state). The result indicates the excellent flexibility and robustness of CF@TiO₂@MoS₂ based fibrous DSSCs.

Fibrous CF@TiO₂@MoS₂ electrode is also applicable in wire-shaped supercapacitors. Figure 4a and Figure S13 of the Supporting Information show the schematic illustration and typical SEM image of solid-state symmetric fibrous supercapacitor based on CF@TiO₂@MoS₂ electrodes and polyvinyl alcohol (PVA)/H₃PO₄ gel electrolyte, respectively. As shown in Figure 4b, the CF@TiO₂@MoS₂ based fibrous supercapacitor displays the highest linear current density, suggesting it has superior energy storage capacity than pristine CF and CF@TiO₂ electrodes. The CV curves of CF@TiO₂@MoS₂ based fibrous supercapacitor at different scan rates were measured (Figure 4c). The shape of these CV curves is approximately rectangular, which indicates that CF@TiO₂@MoS₂ electrodes possess good electrochemical performances during charge/discharge processes. The galvanostatic charge/discharge curves of CF@TiO₂@MoS₂ based supercapacitors at different currents are nearly symmetric straight lines with similar shape (Figure 4d), indicating it is a typical double-layer supercapacitor which can be stably operated in a wide range of charging/discharging currents. The linear

capacitances (C_L) and areal capacitances (C_A) were calculated as well (Figure 4e). The values of C_L and C_A are $1740 \mu\text{F cm}^{-1}$ and 18.51 mF cm^{-2} at the current of 0.02 mA , which are remarkable when comparing to the previous results, as shown in Table S2 of the Supporting Information.^[30–41] Notably, its capacitance retention was about 73% when the charging/discharging current increased from 0.02 to 0.10 mA , showing excellent rate capability. It is known that energy density and power density are two important parameters to evaluate the performances of supercapacitors. Figure S14 of the Supporting Information displays the Ragone plot of CF@TiO₂@MoS₂ based fibrous supercapacitor compared with previously reported fibrous supercapacitors.^[33,35,36,44,70–72] The areal energy density of the device decreases from 1.65 to $1.21 \mu\text{Wh cm}^{-2}$ as the power densities increases from 56.84 to $296.33 \mu\text{W cm}^{-2}$, which are also remarkable when comparing to the previous results.^[33,35,36,44,70–72] Moreover, the cyclic performance was examined by galvanostatic charge/discharge cycling at a current of 0.08 mA (Figure 4f). The fibrous supercapacitor still delivers $\approx 81\%$ of its initial capacitance after 3000 cycles, confirming its good stability.

Electrochemical impedance spectroscopic (EIS) measurements were also performed on CF@TiO₂@MoS₂ based fibrous supercapacitors (Figure S15a, Supporting Information). The vertical line at low frequency and the low equivalent series resistance ($\approx 100 \Omega$) indicate an ideal double-layer capacitive behavior. The flexibility and capacitance retention of CF@TiO₂@MoS₂ based fibrous supercapacitors at different bending angles was also measured (Figure S15b, Supporting Information), exhibiting a retention of 97% as the bending angle changes from 0° to 180° . This result implies the CF@TiO₂@MoS₂ based fibrous supercapacitors possess high stability toward bending. Figure 4g shows a schematic diagram of an integrated high-voltage supercapacitor consisted of three CF@TiO₂@MoS₂ based fibrous supercapacitors connected in series. In comparison with the 0.80 V operating voltage of an individual fibrous supercapacitor, the integrated high-voltage supercapacitor exhibits a broad voltage window of 2.40 V (Figure 4h,i).

In order to confirm that the CF@TiO₂@MoS₂ electrode can be simultaneously used for photoelectric conversion and electrochemical energy storage, an integrated self-powering energy fiber containing an energy harvesting part (fibrous DSSC) and an energy storage part (fibrous supercapacitor) has been fabricated, as illustrated in Figure 5a. In this energy fiber, a CF@TiO₂@MoS₂ electrode acts as both the counter electrode of DSSC part and the cathode of supercapacitor part. The working electrode of DSSC part is dye-sensitized TiO₂ nanotube array grown on Ti wire, while the anode of supercapacitor

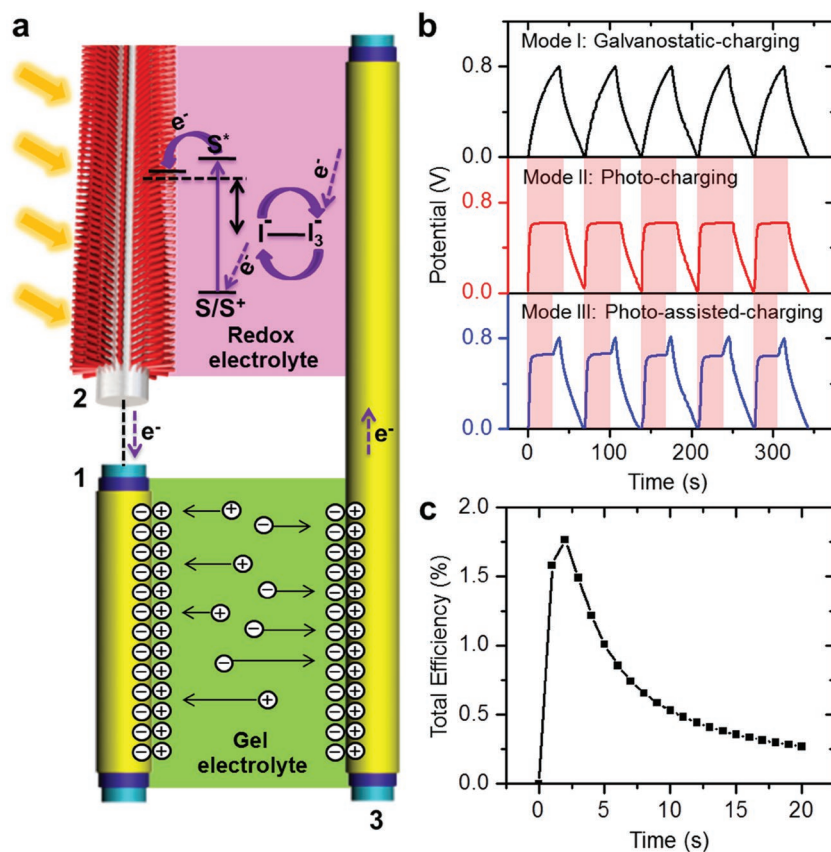


Figure 5. Performances of CF@TiO₂@MoS₂ based self-powering energy fiber. a) Schematic illustration and operational mechanism of a self-powering energy fiber constructed by integrating a fibrous DSSC and a fibrous supercapacitor with CF@TiO₂@MoS₂ electrodes. b) Galvanostatic-charging/galvanostatic-discharging (Mode I), photocharging/galvanostatic-discharging (Mode II) and photoassisted-charging/galvanostatic-discharging (Mode III) curves of the as-prepared self-powering energy fiber, respectively. The time periods of energy fiber under simulated AM1.5G (1 sun) light illumination are marked by light red areas. c) Overall photochemical-electricity energy conversion efficiency of the self-powering energy fiber as a function of the photocharging time.

part is another CF@TiO₂@MoS₂ electrode. Figure 5a displays the charge transport route in the self-powering energy fiber during the photocharging stage. To achieve photocharging of the energy fiber under simulated AM1.5G solar irradiation (1 sun), the working electrode of DSSC part and the anode of supercapacitor part were connected during the charging stages. Upon illumination, dye molecules (N719) were shifted from ground state (S) to excited state (S*), and then the excited electrons were injected into the conduction band of TiO₂, leading to the oxidation of dye molecules (S*→S⁺). At the following step, electrons were transported from the working electrode of DSSC to the anode of the supercapacitor. Meanwhile, the dye molecules returned to the ground state by receiving electrons generated from the oxidation of I⁻ ions to I₃⁻ ions. Then the I₃⁻ ions were diffused to the surface of CF@TiO₂@MoS₂ counter electrode and reduced to I⁻ ions by the catalysis of MoS₂ nanofilms. In this way, continuous light illumination on the DSSC part can achieve the transfer of electrons from the working electrode of the DSSC part to the anode of supercapacitor part, resulting in the fast charging of supercapacitor

part. When the voltage between cathode and anode is close to the V_{OC} , which is the potential between the conduction band of TiO_2 and the redox potential of the I^-/I_3^- electrolyte, the photocharging process will stop. Through this process, the self-powering energy fiber converts solar energy into electric energy by the DSSC part and then stores energy in the supercapacitor part. The stored electrochemical energy can be released by galvanostatic discharging, which is similar to that of conventional supercapacitors.

To investigate the charge/discharge behaviors of the self-powering energy fiber under different conditions, three kinds of testing modes, including galvanostatic-charging/galvanostatic-discharging (Mode I), photocharging/galvanostatic-discharging (Mode II), and photoassisted-charging/galvanostatic-discharging (Mode III) were employed, as detailed in the Experimental Section of Supporting Information. Figure 5b exhibits the corresponding charge/discharge curves of Modes I (black), Mode II (red), and Mode III (blue), respectively. At the galvanostatic charging stage of Mode I (without illumination), the supercapacitor needs ≈ 37 s to reach the cutoff voltage of 0.8 V by applying a constant charging current of 0.06 mA. While under the Mode II, the voltage of supercapacitor can approach the V_{OC} (about 0.65–0.70 V) in just 7 s during the photocharging process under simulated solar illumination (marked by light red color), which is much faster than that under Mode I. Similarly, under Mode III, the supercapacitor only need ≈ 8 s to reach near the V_{OC} under illumination, which further demonstrates the photocharging is fast and effective as compared to the galvanostatic charging under a moderate charging current of 0.06 mA. Then, the voltage of supercapacitor can be raised to the cutoff voltage of 0.8 V by a subsequent galvanostatic charging step. It should be noted that, the voltage plateau during the photocharging process is slightly lower than the V_{OC} , which can be ascribed to the internal resistance of the self-powering energy fiber. Moreover, the conclusion that photocharging process is fast and effective compared to normal galvanostatic charging process is mainly summarized from the performance differences shown in Figure 3g, Figure 4e, and Figure 5b. The fast process of photocharging is mainly resulted from the high current obtained from the DSSCs. As shown in Figure 3g, the V_{OC} , J_{SC} , FF, and η of the $CF@TiO_2@MoS_2$ based fibrous DSSCs are 0.74 V, 16.95 mA cm^{-2} , 0.75, and 9.5%, respectively. Typically, the fibrous DSSC part of energy fiber has a length of 3 cm (with an effective area of ≈ 0.078 cm^2). Based on these data, the largest current generated by DSSCs can be estimated to be around 1.3 mA, which is much higher than that of normal charging/discharging current (0.02 to 0.10 mA, Figure 4d,e). Thus, the photocharging process is fast and effective as compared to normal galvanostatic charging process.

The overall photochemical-electricity energy conversion efficiency of as-prepared energy fiber was calculated using the following equation: $\eta = C \cdot V^2 / (2 \cdot P_{in} \cdot S \cdot t)$.^[29–33] Here C and V are the capacitance and voltage of the energy storage part (fibrous supercapacitor), respectively; P_{in} is the energy density of incident light, S is the effective area of the energy harvesting part (fibrous DSSC), and t is the charging time. According to the photocharging/galvanostatic-discharging curve (Mode II) in Figure 5b, the overall photochemical-electricity energy conversion efficiency as a function of photocharging time is plotted in

Figure 5c. As the photocharging time passes, the overall photochemical-electricity energy conversion efficiency first increases and then decreases (after 2.0 s), showing a maximum value of 1.8%, which is higher than the efficiency of other energy fibers in previous reports (Table S3, Supporting Information).^[39,49–59]

In addition, we have found that $CF@TiO_2@MoS_2$ electrode can also be used in fibrous LIBs. Figure S16a of the Supporting Information shows the schematic diagram of a fibrous LIB consisted of a lithium metal wire anode and a fibrous $CF@TiO_2@MoS_2$ cathode. This fibrous LIB exhibits a high initial working voltage of 3.271 V (Figure S16b, Supporting Information) and can be used as a power source to light up a commercial red LED for about 5 min, as shown in Figure S16c of the Supporting Information. Figure S16d of the Supporting Information displays representative charge/discharge profiles of $CF@TiO_2@MoS_2$ based fibrous LIB measured between 0.01 and 3.0 V at a current rate of 3.0 mA m^{-1} . Specifically, in the first cycle, the fibrous LIB shows discharge and charge capacities of 71.3 and 40.1 mA h m^{-1} , respectively. The initial Coulombic efficiency is 56.2%, which can be ascribed to the formation of solid-electrolyte interface (SEI) films and the partial decomposition of the electrolyte.^[73] The charge/discharge profiles of the subsequent cycles are different with the initial cycle (Figure S16d, Supporting Information). The discharge and charge capacities in the second cycle are 31.4 and 25.5 mA h m^{-1} , respectively, giving a higher Coulombic efficiency of 81.2%. Compared to those values of the second cycle, the discharge and charge capacities are retained as high as 76.1% (23.9 mA h m^{-1}) and 81.0% (20.6 mA h m^{-1}) after 20 charge–discharge cycles, with a Coulombic efficiency of 86.2%. These results suggest that in addition to fibrous DSSCs and supercapacitors, the $CF@TiO_2@MoS_2$ electrode also has the potential to be used in fibrous LIBs.

The HER catalytic activities were measured using the standard three-electrode electrochemical cell setup in 0.50 M H_2SO_4 .^[74–76] The pristine CF, $CF@TiO_2$ and $CF@TiO_2@MoS_2$ electrodes were employed as working electrode, respectively. The counter and reference electrodes were graphite rod and Ag/AgCl electrode, respectively. All of the experimental data have been iR -corrected by subtracting the Ohmic resistance loss from the overpotential. Figure S17a of the Supporting Information shows the Nyquist plots of the three working electrodes, which reveals the Ohmic resistances (R_s) of pristine CF, $CF@TiO_2$, and $CF@TiO_2@MoS_2$ electrodes are 9, 27, and 13 Ω , respectively. The pristine CF electrode displays the lowest R_s , while $CF@TiO_2$ shows the highest R_s , which can be ascribed to the high conductivity of CF and wide-bandgap of TiO_2 . In Nyquist plots, the semicircle in high-frequency region is attributed to the charge transfer at the interface of working electrode/electrolyte, usually associating with R_{ct} . Because of the high conductivity, the R_{ct} of pristine CF electrode is almost zero. The $CF@TiO_2@MoS_2$ electrode reveals a lower R_{ct} value (3.1 Ω) than $CF@TiO_2$ (4.4 Ω), which can be attributed to the larger surface area and higher catalytic activity of $CF@TiO_2@MoS_2$. Figure S17b of the Supporting Information displays the polarization curves of the pristine CF, $CF@TiO_2$, and $CF@TiO_2@MoS_2$ electrodes, respectively, which were obtained at a voltage scanning rate of 5 mV s^{-1} . The $CF@TiO_2@MoS_2$ electrode exhibits much higher activity than the pristine CF and

CF@TiO₂ electrodes. This can be ascribed to the high catalytic activity and large surface area of MoS₂, comparable to the previous reports.^[53,54] Specifically, the onset overpotentials of pristine CF, CF@TiO₂, and CF@TiO₂@MoS₂ electrodes are 454, 124, and 10 mV, respectively. Figure S17c of the Supporting Information shows the corresponding Tafel plots of the pristine CF, CF@TiO₂, and CF@TiO₂@MoS₂ electrodes, respectively. The linear portions of the Tafel slopes are fit to the Tafel equation ($\eta = b \log(j) + a$, where b is the Tafel slope), yielding Tafel slopes of 130.8, 123.7, and 58.4 mV dec⁻¹, respectively. Clearly, the CF@TiO₂@MoS₂ electrode displays the smallest Tafel slope among the three working electrodes, indicating that the CF@TiO₂@MoS₂ electrode can prompt the largest catalytic current density at the same overpotential. This result shows the CF@TiO₂@MoS₂ electrode is a promising catalyst for HER. To evaluate the stability of CF@TiO₂@MoS₂ electrode during electrocatalytic hydrogen evolution in 0.50 M H₂SO₄, the polarization curves before and after 500 cycles CV sweeps are shown in Figure S17d of the Supporting Information. The polarization curve after 500 CV sweeps almost overlays with the initial curve, confirming that the CF@TiO₂@MoS₂ electrode possesses excellent stability in acidic solution.

3. Conclusion

In summary, an all-purpose fibrous CF@TiO₂@MoS₂ electrode was fabricated, and its performances in versatile energy harvesting and storage applications were tested. The fibrous DSSCs and supercapacitors constructed from this novel electrode exhibited superior performances than those in the literature. In addition, an integrated self-powering energy fiber was developed by combining a DSSC part and a supercapacitor part sharing a single fibrous CF@TiO₂@MoS₂ electrode. As a result, this energy fiber exhibited fast charging capability and remarkable energy efficiency. We also found that the CF@TiO₂@MoS₂ electrode can be used for fibrous LIBs and electrocatalytic HER. These applications, especially the integrated self-powering energy fiber, prove that the MoS₂-based fibrous electrode has great potential for the construction of novel and highly efficient power sources to support wearable and portable electronic devices.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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