

Bacteria-Affinity 3D Macroporous Graphene/MWCNTs/Fe₃O₄ Foams for High-Performance Microbial Fuel Cells

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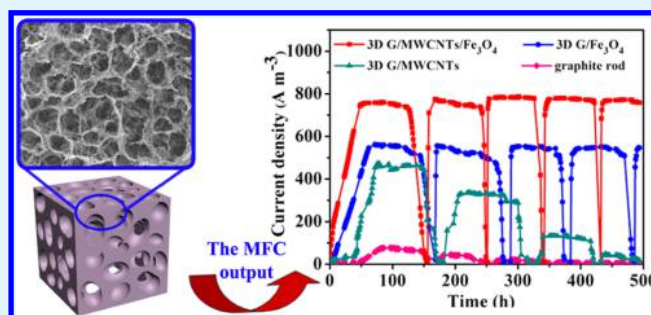
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S Supporting Information

ABSTRACT: Promoting the performance of microbial fuel cells (MFCs) relies heavily on the structure design and composition tailoring of electrode materials. In this work, three-dimensional (3D) macroporous graphene foams incorporated with intercalated spacer of multiwalled carbon nanotubes (MWCNTs) and bacterial anchor of Fe₃O₄ nanospheres (named as G/MWCNTs/Fe₃O₄ foams) were first synthesized and used as anodes for *Shewanella*-inoculated microbial fuel cells (MFCs). Thanks to the macroporous structure of 3D graphene foams, the expanded electrode surface by MWCNTs spacing, as well as the high affinity of Fe₃O₄ nanospheres toward *Shewanella oneidensis* MR-1, the anode exhibited high bacterial loading capability. In addition to spacing graphene nanosheets for accommodating bacterial cells, MWCNTs paved a smoother way for electron transport in the electrode substrate of MFCs. Meanwhile, the embedded bioaffinity Fe₃O₄ nanospheres capable of preserving the bacterial metabolic activity provided guarantee for the long-term durability of the MFCs. With these merits, the constructed MFC possessed significantly higher power output and stronger stability than that with conventional graphite rod anode.

KEYWORDS: microbial fuel cells, bioaffinity, electrochemistry, graphene foams, multiwalled carbon nanotubes



INTRODUCTION

As a classical and widely studied example of bioelectrochemical systems (BESs), microbial fuel cells (MFCs) can fulfill the double task of wastewater treatment and electricity generation, having attracted much attention in recent years.^{1–4} In MFCs, electroactive bacteria are utilized as electrocatalysts to oxidize organic molecules, directly converting chemical energy into electrical energy.^{5–9} However, the low power density and poor long-term stability extremely hinder their practical application. Among numerous factors,^{10–12} the anode material undertakes the function of providing area for bacterial attachment and extracellular electron transfer (EET), highly influencing the overall performance of MFCs. Therefore, extensive studies have been dedicated to developing novel anode materials for overcoming the aforementioned drawbacks.

Generally, commercially available graphite/carbon-based anodes^{13–16} (e.g., graphite rod and carbon paper) are adopted as the anodic matrix in most MFCs. However, these materials inherently suffer from low EET efficiency and poor bacterial adhesion. Moreover, bacterial attachment and electron transfer only occur on the outer surface of materials, while the interior

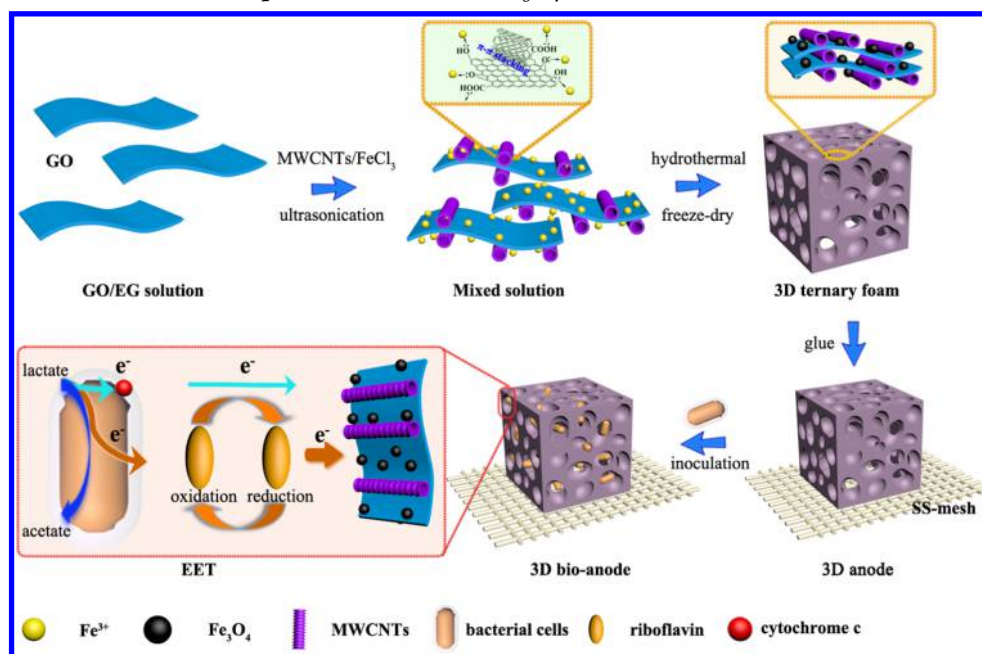
space of these electrodes remains unexploited, owing to the inner pore plugging caused by rapid bacterial growth. In view of this, many endeavors have been made to pursue three-dimensional (3D) materials for configuring high-performance MFC.^{17–24} Among them, 3D graphene emerges as an attractive candidate, due to its outstanding electrical conductivity, large pore size, and high specific surface area.^{25–27} However, during fabrication and drying process, 3D graphene irreversibly aggregates with π - π stacking between graphene sheets (GS),²⁸ dramatically reducing the accessible surface area for bacterial adhesion. To solve this problem, one feasible solution is to incorporate multiwalled carbon nanotubes (MWCNTs), which have been demonstrated to be effective in preventing GS from restacking and accelerating electron transfer.^{29,30}

Aside from the structure optimization of electrode materials for maximized active surface area, regulating electrode composition to improve the bacteria-electrode interactions is

Received: March 21, 2016

Accepted: June 7, 2016

Published: June 7, 2016

Scheme 1. Fabrication of the 3D Macroporous G/MWCNTs/Fe₃O₄ Foams and Their Use as MFC Anode Materials

another effective route for upgrading the MFC overall performance.³¹ The genus *Shewanella*, which consists of dissimilatory iron-reducing bacteria (DIRB) often used as model bacteria in MFCs due to its excellent electrogenic capacity, has the ability to recognize the surface of Fe (III) oxides and initiate EET in its metabolism.^{32,33} The c-type cytochromes on outer-membrane redox proteins (OM c-cyts) mediate EET from bacterial cells to Fe (III) oxides, which act as a “molecular wire” and play an important role in the electron transfer process.^{34–36} Inspired by this mechanism, we deduce that Fe (III) species have the likelihood to favorably attract bacteria and accelerate bacteria-electrode electron transport, thus contributive to elevating the MFC properties at the level of composition modulation other than morphology tailoring.

In this study, adopting MWCNTs as intercalated spacer and Fe₃O₄ nanospheres as bioaffinity anchor, we devise a novel interconnected 3D macroporous framework of graphene sheets incorporated with Fe₃O₄ nanospheres and MWCNTs (3D G/MWCNTs/Fe₃O₄ foams) via one-pot solvothermal process and freeze-dry method (Scheme 1). The resultant 3D G/MWCNTs/Fe₃O₄ foams were employed as an anode for MFCs, displaying much higher power density and better durability compared with a commercial graphite rod anode. The improved performance of MFCs with 3D G/MWCNTs/Fe₃O₄ anode could be attributed to its macropores structure, good conductivity, high affinity for the attachment of *Shewanella*, and efficient EET between microbial biofilm and anode.

EXPERIMENTAL SECTION

Chemicals and Materials. Graphite (ks-10, 99.95%) was purchased from Sigma-Aldrich. Iron(III) chloride hexahydrate (FeCl₃·6H₂O), sodium acetate (NaAc), and ethylene glycol (EG) were obtained from Sinopharm Chemical Reagent Co., Ltd. BCA Multiwalled carbon nanotubes were bought from Nanoport Co., Ltd. (Shenzhen, China). All the other reagents were of analytical grade and used as received. Deionized water (resistance over 18 MΩ cm at 25 °C) used in all experiments was prepared by an ultrapure water system.

3D G/MWCNTs/Fe₃O₄ Foams Preparation. GO was synthesized from natural graphite powder (ks-10, 99.95%) according to a modified Hummer’s method.^{37,38} 3D G/MWCNTs/Fe₃O₄ foams were fabricated by hydrothermal and cryogenic methodologies. In a typical experiment, 0.06 g of MWCNTs and 0.1 g of FeCl₃·6H₂O were added into 10 mL of GO/EG (3 mg mL⁻¹) solution; after ultrasonication for 2 h in an ice bath, NaAc (0.38 g) was added to the solution, followed by vigorous stirring for 15 min to obtain stable mixed suspensions. Two milliliters of the suspensions were transferred to a vial, and then the vial was sealed in a Teflon-lined stainless steel autoclave and maintained at 250 °C for 36 h. During the hydrothermal process, along with the coupling of GO with MWCNTs, the reduction of Fe³⁺ cations to Fe₃O₄ nanospheres occurred in the presence of ethylene glycol (EG) as a reducing agent and sodium acetate as an alkali source. After it cooled, the as-prepared hybrid architecture was first dialyzed with ultrapure water (the ultrapure water was changed every 12 h) for 72 h, followed by freeze-drying to obtain 3D macroporous G/MWCNT/Fe₃O₄ hybrid foams. The mass ratio in this hybrid foam is G/MWCNTs/Fe₃O₄ = 1:2:1. To prepare 3D G/Fe₃O₄ foams and 3D G/MWCNTs foams for comparison, the same conditions and procedures were employed in the absence of MWCNTs and FeCl₃·6H₂O, respectively.

Electrodes Preparation. For 3D electrodes preparation, two monolithic foams (13 mm diameter, 2.5 mm height) were glued to both sides of a stainless-steel (SS, 60 × 60 mesh, type SUS304, Xinsilu Metal Product Co. Ltd., Shanghai, China) mesh.³⁹ The glue was conductive carbon paint (Structure Probe, Inc., SPI Supplies, USA). The SS mesh (15 mm × 20 mm) was placed in acetone for 3 h before using. The total volume of 3D electrodes was 1.3 cm³. To keep the volume of graphite rod electrodes was the same as that of the prepared 3D electrodes, commercial available graphite rods (10 mm diameter, 10 cm height) were cut into slices (10 mm diameter, 4 mm height), which were directly glued to both sides of a SS. Prior to be glued, the slices were sterilized in boiling 0.1 M H₂SO₄ for 30 min and washed several times with ultrapure water, followed by placing in absolute ethanol overnight.

Apparatus. X-ray diffraction measurements were performed on a diffractometer (XRD-6000, Shimadzu, Japan) with Cu Kα radiation (λ = 1.5405 Å). The morphology of the products was observed by a Hitachi S4800 field-emission scanning electron microscopy (FESEM) and a JEOL JEM 200CX transmission electron microscopy (TEM, 200 kV). The Brunauer–Emmett–Teller (BET) specific surface area of the products was obtained using an automated sorption analyzer (ASA

2020M, Micromeritics, USA). The X-ray photoelectron spectra (XPS) were taken on an electron energy spectrometer (K-alpha, Thermo Scientific, USA) using Al K α (1486.6 eV) as the X-ray excitation source. Electrochemical impedance spectroscopy (EIS) was recorded on an Autolab PGSTAT12 (Eco chemie, BV, The Netherlands). The used amplitude was ± 5 mV, and the frequency range was from 1×10^{-2} to 1×10^5 Hz. Cyclic voltammetry analyses were performed using an electrochemical workstation (CHI660D, Chenhua, China) with a conventional three-electrode system.

Bacterial Strain. *S. oneidensis* was inoculated in Luria–Bertani (LB) broth medium and incubated aerobically in a shaking flask (150 rpm, 37 °C) until the optical density (OD₆₀₀) reached ~ 2.4 . The cells for using were harvested by centrifugation (6000 rpm, 5 min) and redispersed in M9 buffer that was composed of 22 mM KH₂PO₄, 42 mM Na₂HPO₄, 85.5 mM NaCl, and 1.0 mM MgSO₄. The M9 buffer was sparged with N₂ to remove oxygen before using. All media were sterilized before using.

Microbial Fuel Cells Construction. Universal H-shaped two-chamber MFCs were constructed as described previously.⁴⁰ The two chambers, anodic and cathodic, were separated by a proton exchange membrane (Nafion 211, Dupont Co). The cathode was a carbon paper (2 cm \times 3 cm, HeSen Electrical Corporation, Shanghai, China), which was connected to a resistance box as well as anode (graphite rod, 3D G/Fe₃O₄ foams, 3D G/MWCNTs foams, or 3D G/MWCNT/Fe₃O₄ foams electrodes) with titanium wire. The anodic chamber was filled with 100 mL of bacterial cell-contained M9 buffer solution with 5% LB broth, and the cathode chamber contained 100 mL of phosphate buffer (100 mM, pH 7.4) and K₃Fe(CN)₆ (50 mM) solution. The MFCs started with adding lactate (1.5 mL) to anodic chamber, and the voltages across a 500 Ω external resistor were recorded. Additional lactate was added when the operating voltage dropped below 0.05 V. The polarization and power-density curves were obtained by varying the external resistance (9000–100 Ω) and using a multimeter to measure the cell voltage. The MFCs were operated at 25 °C with an anaerobic atmosphere.

Scanning Electron Microscopy Analysis. Electrodes with biofilm were removed from MFCs and subsequently cut into several pieces. After fixed in 2.5% glutaraldehyde for 2 h, the pieces were washed with M9 buffer solution and then dehydrated in increasing concentrations of ethanol solutions (25%, 50%, 75%, 85%, 95%, and 100%). Finally, they were dried in a vacuum oven (2 h, 40 °C). The internal and external surfaces of the dried pieces were coated with a thin layer of gold metal using an automated sputter coater (2 Kv, Emitech K550). Then, they were examined with a field emission SEM (S4800 Hitachi, Japan).

Phospholipids Analysis for Determining Microbial Biomass. Electrodes with biofilm were removed from MFCs and analyzed according to the procedure reported previously.^{41–43} The absorbance at 610 nm was determined using a spectro-photometer (3600 Shimadzu, Japan). The concentrations of phosphate were calculated by using the regression line from a standard curve. The total active biomass of electrodes was obtained by using the conversion factor of 191.7 μg of biomass-C per 100 mmol of phospholipid.⁴⁴

Bacterial Cells Viability Analysis. H-shaped two-chamber MFCs equipped with two 3D G/MWCNT/Fe₃O₄ foams were constructed. The foams were separately removed in 100 and 500 h after operation. For comparison, identical MFCs with two 3D G/MWCNT foams were constructed, and the foams were removed with the same procedure as earlier. The viability of bacteria in the foams removed from MFCs was visualized through a Nikon Ti-E inverted fluorescent microscope equipped with a high-resolution CCD camera (DS-Ri1, Nikon Corporation, Japan). The 3D electrodes were cut into small pieces (~ 1 mm height) using a scalpel, and the pieces were then immersed into 20 mL of phosphate buffer (10 mM, pH 7.4) containing 1 mL of calcein-AM fluorescent dyes (2 μM , Keygen Biotech Co. Ltd., Nanjing, China) for 30 min. To remove unbound residual dyes, the pieces were immersed into phosphate buffer for 10 min and repeated three times. The excitation wavelength was 490 nm, and a BP500–550 emission filter was used for green fluorescence.

RESULTS AND DISCUSSION

Characterization of 3D G/MWCNTs/Fe₃O₄ Foams. The fabrication of 3D G/MWCNTs/Fe₃O₄ foams is illustrated in Scheme 1. Graphene oxide (GO) was coupled with MWCNTs through π – π stacking between aromatic rings in their carbon skeletons,⁴⁵ while Fe³⁺ cations were preassembled on the negatively charged GO surface due to the abundant surface oxygen-containing functional groups.⁴⁶ During the hydrothermal process, GO colloids were reduced and shrank into a 3D network while enwrapping MWCNTs and resultant Fe₃O₄ nanospheres in a monolithic manner. The obtained 3D G/MWCNT/Fe₃O₄ foams displayed a column-shaped macrostructure with a diameter of ~ 1.3 cm and a height of 0.25 cm (Figure S1). The morphology of as-synthesized 3D G/MWCNTs/Fe₃O₄ foams was investigated by FESEM and TEM. As shown in Figure 1a, interconnected networks create

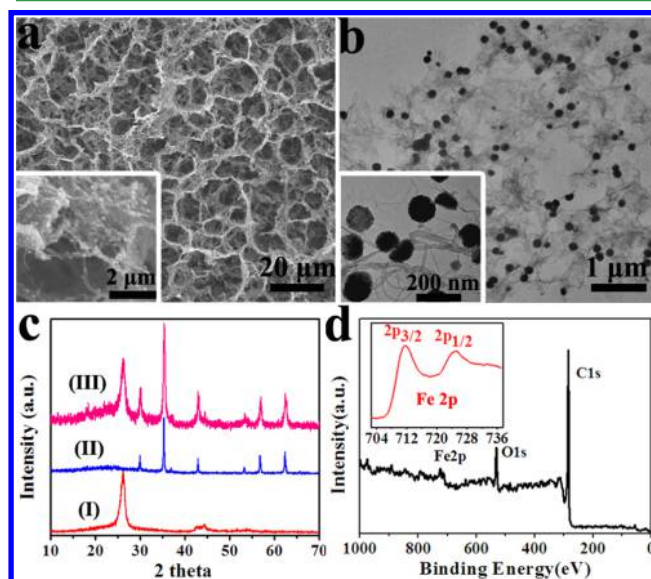


Figure 1. SEM (a) and TEM (b) images of 3D macroporous G/MWCNTs/Fe₃O₄ foams. (insets) The high-magnification images. (c) XRD patterns of MWCNTs (I), Fe₃O₄ (II), and 3D macroporous G/MWCNTs/Fe₃O₄ foams (III). (d) XPS survey spectrum of 3D macroporous G/MWCNTs/Fe₃O₄ foams, the inset corresponding to Fe 2p spectrum.

continuous macropores of ~ 10 μm , which can be beneficial for bacteria (~ 2 μm) penetration into the interior structure.^{25,26,47}

The magnified image shows that Fe₃O₄ nanospheres are attached on both sides of graphene sheets, while MWCNTs bridge the gaps between graphene sheets (Figure 1a inset). TEM image reveals that Fe₃O₄ nanospheres (100–150 nm) and MWCNTs are well-dispersed on graphene sheets (Figure 1b and 1b inset). Moreover, EDX analysis and elemental mapping image confirm the presence of C, O, and Fe elements in the material (Figures S2 and S3), demonstrating the successful formation of G/MWCNTs/Fe₃O₄ ternary composite foams.

The crystal structure of the as-obtained product was examined by XRD analysis (Figure 1c). The peak at $2\theta = 26^\circ$ is observed for MWCNTs,⁴⁸ while the peaks at $2\theta = 30.3^\circ$, 35.7° , 43.3° , 54.1° , 57.4° , 62.9° , corresponding to (2 2 0), (3 1 1) (4 0 0), (4 2 2), (5 1 1), and (4 4 0) are observed for Fe₃O₄.⁴⁹ As for the 3D macroporous G/MWCNTs/Fe₃O₄ foams, all the characteristic peaks of Fe₃O₄ and MWCNTs

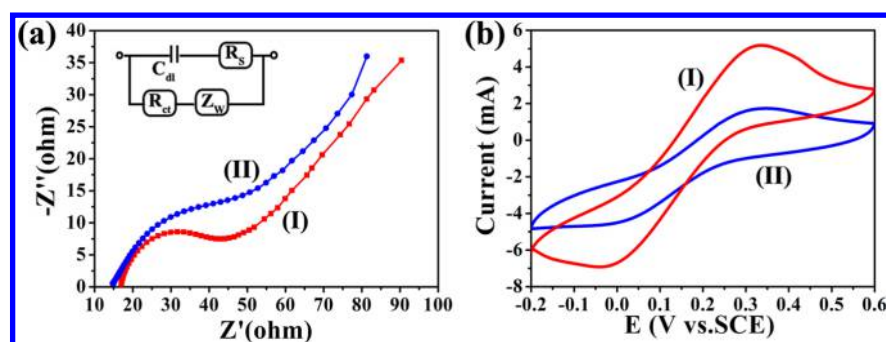


Figure 2. (a) EIS Nyquist plots of different anodes in 2 mM $[\text{Fe}(\text{CN})_6]^{3-/4-}$ and 0.2 M KCl solution, the inset is the Randles equivalent circuit used to fit the EIS data. (b) Cyclic voltammograms of different anodes in 5.0 mM $[\text{Fe}(\text{CN})_6]^{3-/4-}$ and 0.1 M Na_2SO_4 at a scan rate of 100 mV s^{-1} . (I) 3D G/MWCNTs/ Fe_3O_4 electrode, (II) 3D G/ Fe_3O_4 electrode.

are observed, demonstrating the coexistence of Fe_3O_4 and MWCNTs in the composite networks. The XPS was used to further examine the structure of as-prepared G/MWCNTs/ Fe_3O_4 . As shown in Figure 1d, the C 1s, O 1s, and Fe 2p bands are detected in the composite. The two peaks located at 711 and 725 eV in the magnified spectrum are attributed to the Fe 2p_{3/2} and Fe 2p_{1/2} spin orbit peaks of Fe_3O_4 (inset of Figure 1d).⁵⁰ The high-resolution C 1s spectra shows that the intensities of all peaks coming from C–OH, C–O, and C=O (285.6, 286.6, and 288.4 eV) remarkably decrease in comparison with that of GO, indicating that most of the oxygen-containing groups in GO precursor were removed during the one-pot hydrothermal process (Figure S4). In addition, Raman spectra further confirms the reduction of GO (Figure S5). Besides, the peak at 1580 cm^{-1} is designated as D band of MWCNTs,⁴⁸ implying the existence of MWCNTs in the composite.

Electrochemical Properties of 3D G/MWCNTs/ Fe_3O_4 Electrode. To evaluate the contribution of MWCNTs to the ternary composite, 3D G/ Fe_3O_4 foams (see the SEM images in Figure S6a,b) were prepared using a similar procedure as control. On the basis of their N_2 adsorption–desorption isotherms recorded in Figure S7a,b, the BET specific surface area of G/MWCNTs/ Fe_3O_4 foams was calculated to be $168.9 \text{ m}^2 \text{ g}^{-1}$, remarkably higher than that of G/ Fe_3O_4 foams ($83.1 \text{ m}^2 \text{ g}^{-1}$), indicating that the introduction of MWCNTs is effective to inhibit the aggregation of graphene. Apart from the efficacy of spacing graphene nanosheets, MWCNTs could also serve as conductive pathways for electron transport in the electrode substrate,²⁹ which was verified by the decrease of charge transfer resistance (R_{ct}) from 68.9 to 27.5Ω after the addition of MWCNTs (Figure 2a). Moreover, cyclic voltammetry (CV) suggested that the electrochemical active surface area of 3D G/MWCNTs/ Fe_3O_4 anode was much larger than that of the 3D G/ Fe_3O_4 electrode (Figure 2b). As a result of the morphological and electrical merits, the introduction of MWCNTs is supposed to further improve the bacteria-loading capacity of this 3D ternary electrode and enhance EET between bacteria and electrode.

Bacteria-Hosting Capability of 3D G/MWCNTs/ Fe_3O_4 Electrode. To explore the value of Fe_3O_4 for the bacteria-hosting capability of 3D G/MWCNTs/ Fe_3O_4 foams, 3D G/MWCNTs foams (see the SEM images in Figure S6c,d) were also prepared for comparison. After incubation with *Shewanella oneidensis* in typical H-shaped MFCs, the adhesion properties of bacterial cells on those foams were investigated. As observed under SEM (Figure 3a), a multilayered biofilm was formed on

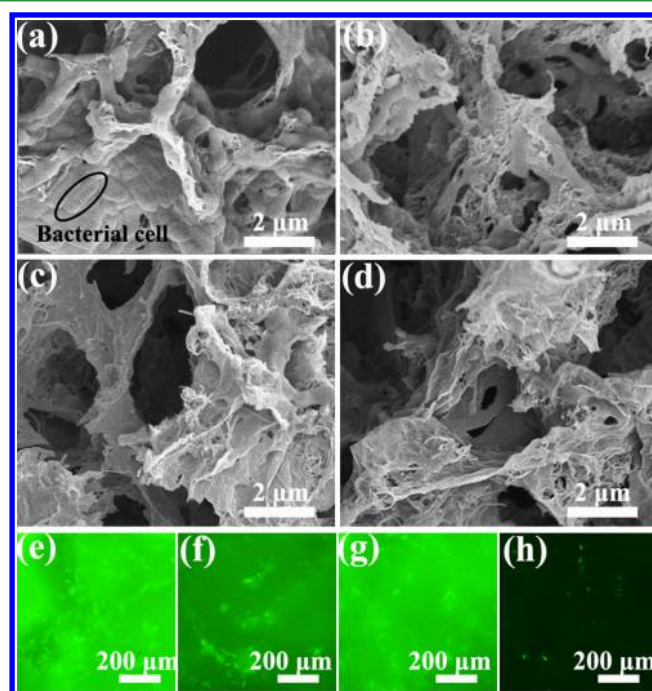


Figure 3. SEM images of *S. oneidensis* MR-1 cells on the surface (a, c) and interior (b, d) of 3D G/MWCNTs/ Fe_3O_4 (a, b), and 3D G/MWCNTs foams (c, d) after 100 h of incubation in MFCs with *S. oneidensis* MR-1 cells. Inverted fluorescent microscopic images of *S. oneidensis* MR-1 cells grown in 3D G/MWCNTs/ Fe_3O_4 (e, g) and 3D G/MWCNTs (f, h) foams for 100 and 500 h, respectively, after staining with calcein-AM fluorescent dye.

the surface of 3D G/MWCNTs/ Fe_3O_4 foams. The SEM images of the electrode after cross-cutting display that bacterial cells also adhere to the inner macroporous architecture of the foams to form a biofilm (Figure 3b). In contrast, in the absence of Fe_3O_4 , both external and internal surface of the 3D G/MWCNTs foams were coated with fewer bacterial cells, leaving large area of substrate uncovered (Figure 3c,d). The total amount of active biomass from 3D G/MWCNTs/ Fe_3O_4 foams was calculated to be $21.69 \text{ mg} \cdot \text{cm}^{-3}$ by the phospholipids analysis, while that of 3D G/MWCNTs foams was $4.34 \text{ mg} \cdot \text{cm}^{-3}$. To better understand the high bacteria-hosting capability of 3D G/MWCNTs/ Fe_3O_4 anode, a comparison of porous structure between 3D G/MWCNTs and 3D G/MWCNTs/ Fe_3O_4 foams was made. On the basis of N_2 adsorption–desorption analysis (Figure S7a,c), the BET specific surface areas of these two foams were approximate ($168.9 \text{ m}^2 \text{ g}^{-1}$ for

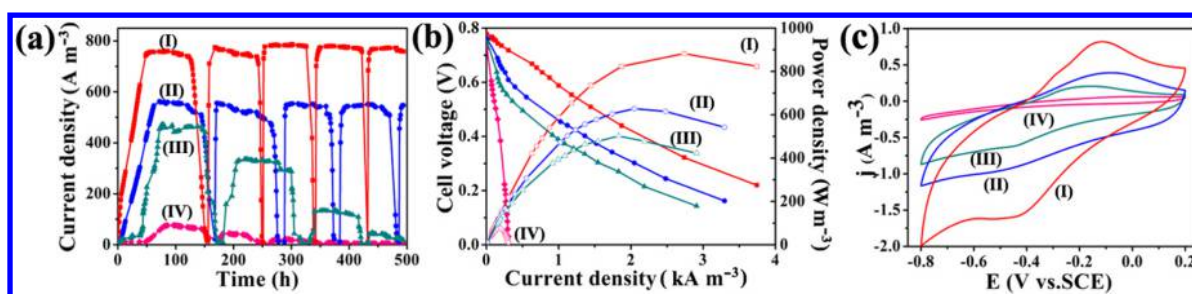


Figure 4. Constant-load discharge curve (a), and polarization and power-density curves (b) of MFCs with different electrodes. (c) CVs of different biofilm-attached anodes of MFCs in fresh mineral medium, scan rate: 10 mV s^{-1} . (I) 3D G/MWCNTs/Fe₃O₄ foams, (II) 3D G/Fe₃O₄ foams, (III) 3D G/MWCNTs foams, and (IV) graphite rod.

3D G/MWCNTs/Fe₃O₄ foams and $142.4 \text{ m}^2 \text{ g}^{-1}$ for 3D G/MWCNTs), and their pore sizes both center at 3.5 nm (Figure S8a,c). Additionally, as estimated from SEM images, the size distribution of micrometer-scale pores in 3D G/MWCNTs foams was similar to that of 3D G/MWCNTs/Fe₃O₄ foams (Figure S9a,c), which suggested that they provided analogous microchannels for bacteria penetration. Considering the similar contribution from their approximate porous structures, the superior bacteria-hosting capability of 3D G/MWCNTs/Fe₃O₄ anode can be mainly attributed to the strong chemotactic behavior of DIRB toward Fe₃O₄ caused by the high affinity between the c-type cytochromes (MtrC and OmcA) on the outer membrane of DIRB and Fe (III) oxide.^{35,36,51}

Moreover, the inverted fluorescent microscope was also used to demonstrate the enhanced metabolic activity of DIRB cells on the 3D G/MWCNTs/Fe₃O₄ foams. After the MFCs operation for 100 and 500 h, the fluorescent microscopic images of bacterial cells grown in the foams were taken. Metabolically active (green) and inactive cells were distinguished using a calcein-AM fluorescent dye based on the differential permeability of cell membranes.⁵² The 3D G/MWCNTs/Fe₃O₄ and G/MWCNTs foams both showed strong green fluorescence after 100 h of operation (Figure 3e,f), implying that a great deal of bacterial cells were alive. Notably, the relatively stronger green fluorescence of 3D G/MWCNTs/Fe₃O₄ foams could be attributed to the increased bacterial attachment compared to 3D G/MWCNTs foams. Interestingly, after 500 h of operation, the fluorescent microscopic image of 3D G/MWCNTs/Fe₃O₄ foams maintained a bright green background (Figure 3g), indicating the sustainable activity of immobilized bacterial cells, in comparison with the fairly weak fluorescence of 3D G/MWCNTs foams (Figure 3h), in which the majority of bacterial cells had been deactivated. In MFCs, microorganisms degrade organic matter, producing electrons that travel through a series of respirator enzymes in the bacterial cell and make energy for bacterial cell in the form of ATP.⁵³ The cytochromes MtrC and OmcA on the outer membrane of DIRB were reported to have a specific affinity with Fe (III) oxide.^{35,36} Such intimate contact could increase the direct electron transfer between bacteria and electrode.^{22,23} Because of the fast release of electrons in bacterial cell, the decomposition rate of organic matter would be accelerated. As a result, bacterial cells could obtain more energy from the decomposition of organic matter for growth and function in the same space of time. We thus concluded that Fe₃O₄ played an important role in keeping the activity of DIRB to support their long-term growth, in addition to the strong affinity toward DIRB.

Performance of Microbial Fuel Cells Configured with 3D G/MWCNTs/Fe₃O₄ Electrode.

With the integration of excellent EET efficiency, high bacterial loading, and enhanced bacterial activity, the MFCs constructed with 3D G/MWCNTs/Fe₃O₄ foams are expected to achieve high performance. To verify this, the performance of MFCs equipped with the 3D ternary foams was tested. As shown in Figure 4a, the volumetric current density of 3D G/MWCNTs/Fe₃O₄ anode increased quickly in the initial 50 h, owing to the fast bacterial cells activation and colonization on anode surface (corresponding time course of the voltage output of the MFCs shows in Figure S10). Then, it kept at a plateau of 772 A m^{-3} for 77 h, followed by a sharp drop due to the depletion of substrate. After replenishment of the substrate, the current density recovered quickly to the level of the first plateau. Even in the fifth cycle, the current density still reached a value as high as 782 A m^{-3} , representing a good long-term stability. Compared to commonly used graphite rod, this 3D anode exhibited an 11-fold increase in biocurrent production ($772 \text{ vs } 67 \text{ A m}^{-3}$), faster starting up (40 vs 85 h), and extremely higher stability. To further clarify the functions of each component in the real practice, the characteristics of MFCs with 3D G/MWCNTs and G/Fe₃O₄ as anodes were assessed in addition. Notably, 3D G/MWCNTs anode possessed starting time (80 h) and degradation behavior similar to graphite rod but starkly different from the ternary composite. This outcome implied that the Fe₃O₄ component took the responsibility of shortening the starting time and increasing life cycles, which should be originated from the enhanced DIRB–anode interaction and the long-term activity of DIRB. However, in the absence of MWCNTs, the maximum current density of 3D G/Fe₃O₄ anode was declined to 562 A m^{-3} , reflecting the importance of MWCNTs on enlarging the electrochemical active surface area and facilitating EET. Moreover, the power density and polarization curves were also tested by varying the external load resistance (Figure 4b). The MFC based on the 3D G/MWCNTs/Fe₃O₄ anode produced a maximum volumetric power density of 882 W m^{-3} , significantly higher than those of graphite rod (76 W m^{-3}) and binary composites (502 and 628 W m^{-3} for 3D G/MWCNTs and G/Fe₃O₄, respectively). Even compared to the previously reported high-output MFCs with 3D anode,^{27,54,55} the proposed MFC with 3D G/MWCNTs/Fe₃O₄ anode possessed comparable output power density. In addition, unlike those in which the current density usually degraded from the third operation cycles, our MFC held stable current density at the high level during five operation cycles, exhibiting its unique merit over the reported 3D anode-equipped MFCs. However, compared to counterpart MFCs with stability level approximate to this work,^{20,56,57} our

proposed MFC produced 1–3 times higher current output. Taken together, this extraordinary character of long-term stability at high output manifests the application potential of this triple-component 3D anode in MFCs.

To get further insight into the influence of the 3D G/MWCNTs/Fe₃O₄ anode on biocurrent generation, the graphite rod, 3D G/MWCNTs, 3D G/Fe₃O₄, and 3D G/MWCNTs/Fe₃O₄ anodes after 100 h of incubation in MFCs were used for CV analysis. As opposed to the flat CV curve of graphite rod anode (Figure 4c, line IV), the 3D G/MWCNTs/Fe₃O₄ anode had one pair of obvious redox peaks at −0.14 and −0.46 V (vs SCE, line I), which were in good accordance with the electrochemical response of the OM c-cyts of *Shewanella*.^{58,59} The result suggested that the 3D G/MWCNTs/Fe₃O₄ anode was able to harvest electrons via direct electron transfer from OM c-cyts, which was inhibited on the conventional graphite rod anode. Meanwhile, the redox peaks of binary composite anodes (3D G/Fe₃O₄, line II and G/MWCNTs, line III) respectively appeared weaker than that of 3D G/MWCNTs/Fe₃O₄ anode, further highlighting the indispensability of both MWCNTs and Fe₃O₄ components in enhancing the direct electron transfer. As reported previously, the use of CNTs could facilitate the electrode to get closer to the active center of OM c-cyts.⁶⁰ Furthermore, the modification of Fe₃O₄ has also been demonstrated to increase the kinetic activity of electrode.⁶¹ Therefore, the introduction of MWCNTs and Fe₃O₄ could be regarded as the essential factors for enhancing direct electron transfer from OM c-cyts to 3D G/MWCNTs/Fe₃O₄ anode.

CONCLUSIONS

In summary, the novel 3D G/MWCNTs/Fe₃O₄ foams have been successfully prepared by one-pot solvothermal method and applied as an anode for MFCs. Besides the enhanced surface area provided by the 3D macroporous structure and the high conductivity of MWCNTs, this anode possesses high bacterial loading capacity and activity-retention ability due to the incorporation of DIRB-affinity Fe₃O₄, all of which significantly contributed to the improved overall performance. By coupling morphology tailoring with composition modulation, this design upgrades both the long-term stability and output performance of MFC, which brings a promising future for the development of MFC anode materials.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsami.6b03425.

Supplementary characterizations of the 3D G/MWCNTs/Fe₃O₄ foams, SEM images of 3D G/MWCNTs and G/Fe₃O₄ foams, and N₂ adsorption–desorption isotherms of 3D G/MWCNT/Fe₃O₄, G/MWCNTs, and G/Fe₃O₄ foams. (PDF)

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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

We gratefully appreciate the support from National Basic Research Program of China (2011CB933502) and the National Natural Science Foundation of China (21175065, 21375059, and 21335004). The authors extend their sincere appreciation to the Deanship of Scientific Research at King Saud Univ. for its funding this prolific Research Group (PRG-1437-32).

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